## Effect of Ytterbium Substitution on the Structural and Magnetic Properties of Nanocrystalline Zinc Ferrite

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In recent years, partial rare earth substitution for iron has been widely used to modify the composition of spinel ferrites. Analysis of published data indicates that the addition of even small amounts of rare-earth metals to ferrites changes their magnetic, electrical, and structural properties. The addition of Dy<sup>3+</sup> cations to a Ni-Cu-Zn ferrite improves the magnetic properties of the material. According to Sattaret al. [6], Yb<sup>3+</sup> cations increase the magnetic permeability of a Cu-Zn ferrite, whereas La<sup>3+</sup> cations were reported to reduce the magnetization and coercive force of a Zn-Cu-Ni ferrite and raise the activation energy for conduction in this material. The enhancement of the magnetic and electrical properties of the spinel ferrites is commonly interpreted in terms of the magnetocrystalline anisotropy of the rare-earth elements and the exchange interaction between the Fe<sup>3+</sup> and Ln<sup>3+</sup> cations. At the same time, attention should be paid to the fact that the reported composition ranges of the forming solid solutions differ significantly.

Rare earth doped Zinc ferrites were prepared  $Zn(Yb_xFe_1, x)_2O_4$  with (x=0.0000, 0.0125, 0.0250, 0.0375 and 0.0500) by the solid state method at 900°C to be single-phase. The cubic cell parameter *a* of the spinel phase was shown to increase linearly from 8.47 Å to 8.55 Å. However, the *a* cell parameter of the samples was found to increase over the entire composition range studied (x=0.0000, 0.0125, 0.0250, 0.0375 and 0.0500). Note that, In the presence of impurity phases, the rate of the increase in the *a* cell parameter of the cubic spinel phase was slower. The formation of a limited

#### ABSTRACT

I.

A series of ytterbium substituted Zinc ferrites Zn (YbxFe1-x)204 with (x=0.0000, 0.0125, 0.0250, 0.0375, 0.0500) was synthesized by the solid state method. The structural characterizations of all the prepared samples were done by using X-ray diffraction (XRD). These studies confirmed the formation of single-phase structure in all compositions. The increase in the value of lattice parameter with increase in ytterbium concentration suggests the expansion of unit cell. Crystallinity and the crystallite size are observed to increase with the concentration of Ytterbium. The substitution of ytterbium strongly influences the magnetic characteristics and this is confirmed from the magnetization measurements at room temperature.

KEYWORDS: ytterbium, XRD, FTIR, magnetic property

### INTRODUCTION

This paper deals with the preparation conditions, composition Ranges, structure, morphological characteristics, and magnetic properties of solid solutions based on rare-earth-doped. $ZnFe_2O_4$  zinc ferrite with the spinel structure. This ferrite is used as a material for electromagnetic radiation absorption and should have high resistivity, low dielectric losses, high hardness, high Curie temperature, and good chemical stability. There are a variety of approaches, including compositional changes, for improving the performance of microwave-absorbing zinc ferrite-based coatings in order to reduce their reflectivity for electromagnetic waves either in a wide frequency range or near one frequency of interest.

solid solution series was also reported for  $YbZnFe_2O_4$  ferrites. The x=0.0000, 0.0125, 0.0250, 0.0375 and 0.0500 material synthesized at 900°C was reported to be single-phase, whereas the materials of this composition prepared at 1100°C, and 1200°C contained ytterbium ferrite, YbFeO\_3.

These findings are consistent synthesis, structure, and magnetic properties with reported results for YbZnFe<sub>2</sub>O<sub>4</sub> ferrites, which demonstrate that the maximum degree of rare-earth substitution for Fe is x = 0.0500. In this paper, we report the preparation of zinc ferrites with the general formula Zn (Yb<sub>x</sub>Fe<sub>1-x</sub>)<sub>2</sub>O<sub>4</sub> (x = 0.0000, 0.0125, 0.0250, 0.0375 and 0.0500), doped with light and heavy lanthanides (Nd, Gd, Yb, and Lu).

The ferrites were obtained by two processes: selfpropagating high-temperature synthesis (SHS) and thermal decomposition of carboxylates (salts of carboxylic acids). We determined the phase boundaries of the corresponding solid solution series and investigated the crystal structure, microstructure, and magnetic properties of ceramic samples.

### II. EXPERIMENTAL PRODUCTURE

The solid state method relies on the heat treatment of a compacted mixture of iron, zinc, and rare-earth nitrates. As starting chemicals for the preparation of mixtures, we used ZnO (analytical grade), Fe (metal), Yb<sub>2</sub>O<sub>3</sub> (Yb = Nd, Gd, Ln, Lu; 99.98%). An appropriate mixture of the starting chemicals was loaded into an Agate mortar and ground for 5 hours. After mixing and grinding, the mixture has been pre

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sintered at 900°C for 5 hours in a furnace with heating rate of 20°C/min and cooled to room temperature with the same rate. After that, the powder and the mixture has been ground with an Agate mortar for 1 hour. Then, the powder has been pressed into pellets and toriods by uniaxial hydraulic press at a pressure of 5 tons.

The sintering temperature should start from 1000°C and 1100°C which is 100 °C higher than the pre-sintering temperature. This condition is required for densification of ferrite in standard solid state methods. Therefore, the final heat treatment step was carried out at temperatures from 1000°C to 1100°C with time duration for 5 hours to densify ZnFe<sub>2</sub>O<sub>4</sub> and Yb doped ZnFe<sub>2</sub>O<sub>4</sub> magnetic ferrite by standard solid state method. The phase composition of the samples thus prepared was determined by X-ray diffraction (Shimadzu XRD-7000S diffractometer, CuKα radiation) and by analyzing energy dispersive X-ray spectra obtained on a JEOL JSM-5900L scanning electron microscope using an EDX detector. Lattice parameters were refined with FullProf-2010 software. The magnetic properties of the ytterbium doped zinc ferrite samples were studied in static magnetic fields of up to 30 kOe (2.4 MA/m) using a soft magnetic Hysteresis Graphs Meter (DX-2012SD).

#### **RESULTS AND DISCUSSION** III.

X-ray diffraction characterization showed that, independent of the synthesis process, the major synthesis product was a Zn  $(Yb_xFe_{1-x})_2O_4$  solid solution with a cubic spinel structure. In all instances, a single-phase material was only obtained at the rare earth content (*x*= 0.0125, 0.0250, 0.0375, 0.0500) in Figure 1. Table 1 summarizes the structural characteristics of the solid solutions obtained. It follows from these data that the cubic cell parameter *a* of the rare earth- doped ferrites varies in a complex manner. The a cell parameter first oper Table 1 Crystallite size of the ZnYbFe<sub>2</sub>O<sub>4</sub> samples at increases and then, as a rule, decreases.

As seen in XRD spectra cubic spinel structure of zinc ferrite was found to be stable at final sintering temperature. The lattice constant and crystallite size of  $Zn(Yb_xFe_{1-x})_2O_4$ calculated data are shown in Table 1. The lattice parameter obtained for pure zinc ferrite is in good agreement with the reported value.

The diffraction peaks corresponding to (2 2 0), (3 1 1), (4 0 0),  $(4\ 2\ 2)$ ,  $(5\ 1\ 1)$  and  $(4\ 4\ 0)$  reflection planes and the absence of any extra peak show that all the samples have attained single phase cubic structure. This implies that the Yb<sup>3+</sup> ions have been completely dissolved into the spinel lattice of zinc ferrite.

The lattice constant of ytterbium-substituted zinc ferrite is observed to be larger than that of zinc ferrite. An increase in lattice constant with increase in Yb3+ ion content is expected because of the large ionic radius of Yb<sup>3+</sup> (0.858 Å) compare to that of  $Fe^{3+}(0.67 \text{ Å})$ .

The ytterbium substitution zinc ferrite changes in the lattice constant and this may probably be the reason for the observed shift of XRD peaks with ytterbium substitution. The theoretical (X-ray) density of zinc ferrite is in agreement with that of the zinc ferrite because it increases with increase in ytterbium concentration.

#### A. Magnetic Hystersis Loop Analysis

The magnetic hystersis loops at room temperature were recorded by Magnetometer (DX-2012SD). The hysteresis loop data of Zn  $(Yb_xFe_{1-x})_2O_4$  (x = 0.0000, 0.0125, 0.0250, 0.0375 and 0.0500) are shown in Table 2 and Table 3. It is clear that by applying magnetic field of 10,000 Oe at room temperature, the saturation magnetization can be achieved for these samples. The value of M<sub>s</sub> is found to be decreased as the amount of doping ions increased. It has been reported extensively that the smaller size particles exhibit the low value of saturation magnetization, which is due to their modified cationic distribution and surface disorder . According to the references, the magnetic moment of Zn<sup>2+</sup> is 0.3  $\mu_B$ , that of Yb^{2+} is 0.85  $\mu_B$  and that of Fe^{2+} is 4.9  $\mu_B$ -6.7  $\mu_B$ . Therefore, the value of M<sub>s</sub> decreased as the amount of Yb doping ions with smaller magnetic moment was increased. According to brown's relation, the coercivity and saturation magnetization are given by the relation [3].

$$H_c = \frac{K_1}{\mu_0 M_s}$$

In equation, the  $K_1$  is magneto-crystalline anisotropy,  $\mu_0$  is vacuum permeability, M<sub>s</sub> is saturation magnetization and H<sub>c</sub> is coercivity. It has been observed that H<sub>c</sub> and M<sub>s</sub> of the prepared materials are inversely related by increasing substitution level and are in agreement with equation <sup>[3]</sup>. The value of H<sub>c</sub> increased with the enhancement of doping ions of Yb<sup>+3</sup>. The low values of coercivity depicted the soft character of these materials which can be considered better candidate for switching, sensing and security applications. These low H<sub>c</sub> values are necessary for electromagnetic (EM) materials. From the hysteresis loop, a number of primary magnetic properties of a ferrite material can be determined.

# different temperatures

| Sample (x) | Temperature<br>1000( ºC) |                     | Temperature<br>1100( ºC) |                     |
|------------|--------------------------|---------------------|--------------------------|---------------------|
|            | Crystallite<br>size (nm) | Lattice<br>Constant | Crystallite<br>size (nm  | Lattice<br>Constant |
|            |                          | (Å)                 |                          | (Å)                 |
|            |                          |                     |                          |                     |
| x = 0.0000 | 44.02                    | 8.47                | 42.46                    | 8.47                |
| x = 0.0125 | 36.78                    | 8.47                | 43.80                    | 8.50                |
| x= 0.0250  | 44.29                    | 8.47                | 44.76                    | 8.49                |
| x = 0.0375 | 41.18                    | 8.48                | 47.30                    | 8.49                |
| x = 0.0500 | 49.91                    | 8.49                | 51.77                    | 8.55                |

#### Table2 Magnetic hysteresis loop data for Zn(YbxFe1x)204 at 1000°C

| Composition of     | Ms      | Нс     | n <sub>B</sub>          |
|--------------------|---------|--------|-------------------------|
| rare earth ions(x) | (emu/g) | (0e)   | <b>(μ</b> <sub>B)</sub> |
| 0.0000             | 94.20   | 92.84  | 3.51                    |
| 0.0125             | 76.5    | 119.26 | 2.83                    |
| 0.0250             | 69.47   | 130.57 | 2.68                    |
| 0.0375             | 67.47   | 143.77 | 2.61                    |
| 0.0500             | 61.99   | 155.04 | 2.57                    |

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#### Table3 Magnetic hysteresis loop data 7n(VhyEq1-y)204 at 1100°C

| Composition of<br>rare earth ions(x) | Ms<br>(emu/g) | Нс<br>(Oe) | n <sub>B</sub><br>(μ <sub>B)</sub> |  |  |  |
|--------------------------------------|---------------|------------|------------------------------------|--|--|--|
| 0.0000                               | 94.20         | 92.84      | 3.51                               |  |  |  |
| 0.0125                               | 76.5          | 119.26     | 2.83                               |  |  |  |
| 0.0250                               | 69.47         | 130.57     | 2.68                               |  |  |  |
| 0.0375                               | 67.47         | 143.77     | 2.61                               |  |  |  |
| 0.0500                               | 61.99         | 155.04     | 2.57                               |  |  |  |



Figure1 XRD pattern of the ZnYbFe<sub>2</sub>O<sub>4</sub> samples at different temperatures



Figure2 Variation of Saturation magnetization with different Yb Content (x) in Zn (Yb<sub>x</sub>Fe<sub>1-x</sub>)<sub>2</sub>O<sub>4</sub> at 1000°C



Figure3 Variation of Coercivity with different Yb Content (x) in Zn (Yb<sub>x</sub>Fe<sub>1x</sub>)<sub>2</sub>O<sub>4</sub> at 1000°C



Figure4 Variation of Saturation magnetization with different Yb Content (x) in Zn (Yb<sub>x</sub>Fe<sub>1-x</sub>) <sub>2</sub>O<sub>4</sub> at 1100°C



Figure5 Variation of Coercivity with different Yb Content (x) in Zn (YbxFe1-x)2O4 at 1100°C

### Conclusion

In this experiment of ytterbium doped on the zinc ferrite Zn  $(Yb_xFe_{1-x})_2O_4$  with x=0.0000, 0.0125, 0.0250, 0.0375 and 0.0500 were synthesized by using solid state method. XRD analysis confirmed the formation of single phase structure, without any secondary phase in all the compositions. The substitution of ytterbium in zinc ferrite has resulted in an increase in lattice constant. Moreover, it was observed that coercivity H<sub>c</sub> increased by increasing doping contents of Yb<sup>3+</sup> ions. It might be due to the effect of denser microstructure with Yb<sup>3+</sup> ions substitution in spinel lattice. This is proved that the substitution of small amount of ytterbium rare earth (RE) ions in ferrite can also tune the magnetic properties.

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